

Fig. 1. Projection de la structure le long de *c*.

**Discussion.** Cette structure est caractérisée par la présence d'une hélice d'atomes d'arsenic parallèle à l'axe *c*. La liaison As—As, la plus courte de la structure, est sensiblement constante (2,42 à 2,43 Å) et comparable à celles observées dans d'autres combinaisons polyanioniques de l'arsenic: 2,43 Å dans CdAs<sub>2</sub> (Horn & Lukaszewicz, 1968) et 2,40 Å dans Cd<sub>4</sub>As<sub>2</sub>I<sub>3</sub> (Gallay, 1975). Ces différentes hélices sont reliées entre elles par les atomes de cadmium.

Chaque atome d'arsenic est entouré par les deux atomes d'arsenic de l'hélice et par deux atomes de cadmium, ces quatre voisins constituant un tétraèdre très déformé.

Chaque atome de cadmium est pentacoordonné, il est au centre d'une bipyramide déformée constituée par trois atomes d'arsenic et par deux atomes d'iode. Les liaisons varient de 2,62 à 2,74 Å pour Cd—As et de 2,98 à 3,43 Å pour Cd—I, elles sont comparables respectivement aux liaisons observées dans CdAs<sub>2</sub> (2,69 Å; Horn & Lukaszewicz, 1968) et CdI<sub>2</sub> (2,99 Å; Wyckoff, 1963).

Chaque atome d'iode est entouré par quatre atomes de cadmium, l'ensemble étant sensiblement coplanaire. Dans les files Cd—I—Cd—I... les angles I—Cd—I valent 167 et 170°.

Il n'y a aucune liaison Cd—Cd ni aucune liaison As—I, les plus courtes distances Cd—Cd et As—I étant respectivement de 3,90 et 3,88 Å. Formellement, le polyanion a une charge *n* fois négative. Si l'on applique la formule classique des composés de valence (Parthé, 1972) on trouve que le nombre de liaisons anion—anion par anion est de 1,5 ce qui correspond bien à deux liaisons anion—anion par arsenic, l'iode ne participant pas à ces liaisons polyanioniques.

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## Tungsten Dioxide: Structure Refinement by Powder Neutron Diffraction

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**Abstract.** WO<sub>2</sub>, monoclinic, space group *P2<sub>1</sub>/c*, *a* = 5.563 (2), *b* = 4.896 (3), *c* = 5.663 (2) Å, β = 120.47 (1)° (298 K). The structure has been refined by profile analysis of powder neutron diffraction data at room temperature. It is a monoclinically distorted rutile structure with alternate W—W lengths of 2.475 and 3.096 Å along the rutile chains.

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**Introduction.** WO<sub>2</sub> exists as a line phase at ambient pressures over the temperature range 273 < *T* < 1803 K (*Phase Diagrams for Ceramists*, 1969). Powder X-ray data show that it is isostructural with MoO<sub>2</sub> (Magnéli & Andersson, 1955).

WO<sub>2</sub> was prepared by reduction of WO<sub>3</sub> (Koch Light Laboratories) in a static H<sub>2</sub>/H<sub>2</sub>O atmosphere, © 1979 International Union of Crystallography

with  $H_2/H_2O$  pressure ratio of 7.7:1, at 1023 K for 24 h. The product, a brown powder, was characterized by its powder X-ray pattern and the composition  $WO_{2.00(1)}$  confirmed by its reducing power (Chaoin & Marion, 1963) and by thermogravimetric reduction.

The neutron powder profile was recorded on the diffractometer PANDA at AERE Harwell in the  $2\theta$  range 10 to  $110^\circ$  in steps of  $0.1^\circ$ . The sample ( $\sim 10$  g) was contained in a thin-walled vanadium can. The powder profile was recorded at room temperature with a neutron wavelength of 1.2615 Å, obtained by reflexion from the (511) planes of a Ge monochromator with a take-off angle of  $70^\circ$ .

$MoO_2$  has been the subject of a single-crystal X-ray investigation (Brandt & Skapski, 1967). The space group is  $P2_1/c$  with specific absences  $0k0$ ,  $k = 2n + 1$  and  $h0l$ ,  $l = 2n + 1$ . Both powder X-ray and powder neutron profiles for  $WO_2$  may be indexed on this scheme, there being no evidence for the space group  $P2_1$ . Thus the method of least-squares profile refinement (Rietveld, 1969) was applied in the space group  $P2_1/c$ , with as starting parameters the atomic coordinates of  $MoO_2$ . This program minimizes the function  $\chi^2 = \sum_i w_i [y_i(\text{obs}) - cy_i(\text{calc})]^2$ , where  $w_i$  is the weighting function,  $y_i$  is the number of counts at the point  $2\theta_i$  and  $c$  is the scale factor. The scattering lengths were  $b_W = 0.48$ ,  $b_O = 0.580 \times 10^{-14}$  m (Bacon, 1972). The refinement included the scale factor, three halfwidth parameters, the zero-point correction, the lattice constants, nine positional parameters, three thermal parameters and the peak asymmetry parameter for peaks below  $2\theta = 40^\circ$ .

$$R_{\text{profile}} = 100 \left\{ \frac{\sum_i w_i [y_i(\text{obs}) - cy_i(\text{calc})]^2}{\sum_i w_i [y_i(\text{obs})]^2} \right\}^{1/2}$$

converged to 9.28 compared with 9.35 expected on purely statistical grounds.  $R_i = 100 \sum |I_{\text{obs}} - cI_{\text{calc}}| / \sum I_{\text{obs}}$  was 6.08. Atom parameters are given in Table 1, and the agreement between observed and calculated profiles is shown in Fig. 1.

**Discussion.** Bond angles and distances are given in Table 2. A projection of the structure on (010) is given in Fig. 2. The bond lengths are derived from the X-ray lattice parameters.

The structural details of  $WO_2$  and  $MoO_2$  are very similar, both being of a monoclinically distorted rutile

Table 1. Atom parameters for  $WO_2$

	x	y	z	B (Å <sup>2</sup> )
W	0.2278 (11)	-0.0102 (14)	0.0111 (12)	0.19 (5)
O(1)	0.1119 (11)	0.2186 (11)	0.2334 (8)	0.31 (7)
O(2)	0.3900 (9)	0.7024 (12)	0.2986 (9)	0.57 (8)

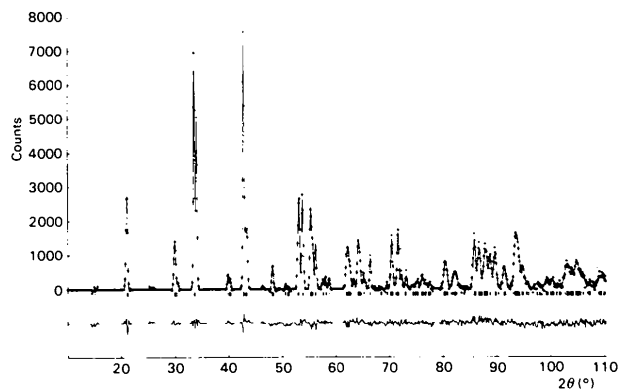


Fig. 1. Observed and calculated profiles for  $WO_2$ : ● observed; — calculated. Upper trace: reflexions; lower trace: difference plot.

Table 2. Interatomic distances and angles in  $WO_2$

(a) Interatomic distances (Å) (primed symbols represent atoms on an adjacent plane)

W(1)—W(2)	2.475 (2)	O1(1)—O1(4)	2.796 (2)
W(1)—W(3)	3.096 (2)	O1(1)—O2(1)'	2.750 (2)
		O1(2)—O2(3)'	2.706 (1)
		O1(1)—O2(4)	2.721 (1)
		O1(1)—O2(5)'	3.445 (1)
		O2(1)—O2(4)	2.792 (1)
Within $WO_6$ octahedron			
W(1)—O1(1)	2.019 (2)	O1(1)—O1(2)	2.848 (1)
W(1)—O1(2)	1.975 (2)	O1(1)—O2(3)	2.868 (2)
W(1)—O1(3)	1.969 (1)	O1(1)—O1(3)	3.128 (1)
W(1)—O2(1)	1.989 (2)	O1(1)—O2(1)	2.887 (2)
W(1)—O2(2)	2.061 (2)	O2(1)—O2(3)	2.912 (2)
W(1)—O2(3)	2.107 (1)	O2(1)—O2(2)	2.870 (2)

(b) Bond angles ( $^\circ$ ) (typical e.s.d. for angles =  $0.1^\circ$ )

W(2)—W(1)—W(3)	173.9		
Within $WO_6$ octahedron			
Around W		Around O(1)	
O1(1)—W(1)—O2(2)	171.8	O1(2)—O1(1)—O2(3)	56.9
O1(2)—W(1)—O2(1)	173.2	O2(3)—O1(1)—O2(1)	60.8
O1(3)—W(1)—O2(3)	167.2	O2(1)—O1(1)—O1(3)	57.8
O1(1)—W(1)—O1(3)	103.3	O1(2)—O1(1)—O2(1)	87.2
O1(1)—W(1)—O1(2)	91.0	Around O(2)	
O1(1)—W(1)—O2(1)	92.2	O1(1)—O2(1)—O2(3)	59.3
O1(1)—W(1)—O2(3)	88.0	O2(2)—O2(1)—O2(3)	57.8
O2(1)—W(1)—O1(3)	94.8	O2(2)—O2(1)—O1(3)	55.8
O2(1)—W(1)—O2(3)	90.6	O2(2)—O2(1)—O1(1)	90.0
O2(1)—W(1)—O2(2)	90.2		
O2(3)—W(1)—O2(2)	84.1		
O2(3)—W(1)—O1(2)	83.5		

type, with the bond lengths in the  $WO_6$  octahedra slightly longer (average W—O = 2.020, average Mo—O = 2.011 Å). This is consistent with the ionic radii of  $W^{4+}$  and  $Mo^{4+}$  (0.66 and 0.650 Å, respectively, Shannon, 1976). The important structural feature is the occurrence of alternately short and long W—W distances down the rutile chains (the  $a$  axis). This has been rationalized in terms of metal—metal bonds (Sleight, Hare & Sleight, 1968) and is also seen in other metal dioxides with rutile-related structures

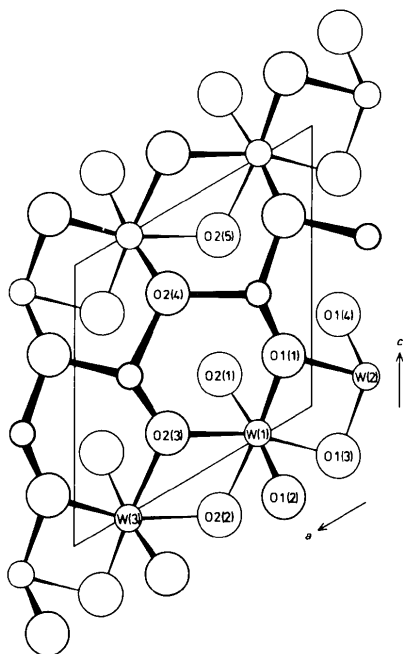


Fig. 2. A projection of the structure of  $\text{WO}_2$  on to the (010) plane.

such as  $\text{VO}_2$ ,  $\text{TcO}_2$  and  $\text{NbO}_2$  (Rogers, Shannon, Sleight & Gillson, 1969) as well as in  $\text{AlWO}_4$  (Doumerc, Vlasse, Pouchard & Hagenmuller, 1975) and  $\text{GaWO}_4$  (Palmer, 1977).

There is no evidence for the space group  $P2_1$  as suggested by Magnéli & Andersson (1955) from either the X-ray or neutron powder profiles: the symmetry

and size of the unit cell of  $\text{WO}_2$  are such that there would be no fully resolved peaks  $h0l$ ,  $l = 2n + 1$  in the  $2\theta$  range investigated. However, refinement was attempted in  $P2_1$ ; no improvement in the agreement between observed and calculated profiles was seen, and indeed the atomic coordinates refined effectively to the symmetry-equivalent positions in  $P2_1/c$ .

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## Dieuropium(III) Germanate Oxide

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**Abstract.**  $\text{Eu}_2(\text{GeO}_4)_2\text{O}$ , monoclinic,  $P2_1/c$ ,  $a = 9.977(7)$ ,  $b = 7.136(4)$ ,  $c = 6.854(4)$  Å,  $\beta = 115.78(6)^\circ$ ,  $Z = 8$ ,  $D_m = 6.84(2)$ ,  $D_x = 6.90$  Mg  $\text{m}^{-3}$ . Crystals were prepared by melting a pre-sintered mixture of  $\text{Eu}_2\text{O}_3$  and  $\text{GeO}_2$  in an image furnace with a tungsten halogen lamp as heat source. The seven-coordination polyhedra of  $\text{Eu}^{3+}$  ions in a set of

equivalent sites form slabs parallel to (100) by edge-sharing. These slabs are connected to the corrugated layers of the edge-shared polyhedra of seven-coordinated  $\text{Eu}^{3+}$  ions in other equivalent positions through common edges of the polyhedra and O–Ge–O bonds of the  $\text{GeO}_4$  tetrahedra.

**Introduction.** Of the germanates and silicates of the rare-earth elements  $\text{Ln}_2\text{XO}_5$  ( $\text{Ln} = \text{Sc}, \text{Y}, \text{La}, \text{Ce}, \dots$ , © 1979 International Union of Crystallography

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